

## Theory behind atomistic simulations

[main source: Allen-Tildesley]

- An atomistic simulation (MD or MC) gives atom positions and velocities {q<sub>i</sub>, p<sub>i</sub>}
- {q<sub>i</sub>, p<sub>i</sub>} (or in cartesian coordinates {r<sub>i</sub>, p<sub>i</sub>}) ⇒ macroscopic quantities (This is what statistical physics is all about!)
- system Hamiltonian H(q, p)
- equations of motion: $\dot{q}_k = \frac{\partial}{\partial p_k} H(\mathbf{q}, \mathbf{p}) \dot{p}_k = -\frac{\partial}{\partial q_k} H$
- N particles  $\Rightarrow$  the system state at any given time is a point  $\Gamma$  in a 6N-dimensional phase space.
  - The evolution of the system from one point  $\Gamma$  to another is determined by the MD equations of motion or a Metropolis Monte Carlo simulation.

### Theory behind atomistic simulations

- One point in phase space  $\{q_i, p_i\} = \Gamma$
- Measured (macroscopic) quantity  $A_{obs}$  corresponding to (microscopic) physical quantity  $A = A(\Gamma)$  from MD simulations as a time average:

$$A_{\text{obs}} = \langle A \rangle_t = \langle A(\Gamma(t)) \rangle_t = \lim_{t_{\text{obs}} \to \infty} \frac{1}{t_{\text{obs}}} \int_0^{t_{\text{obs}}} A(\Gamma(t)) dt$$

• All practical simulations are of course over discrete steps, so the integral has to be rewritten

$$A_{\text{obs}} = \langle A \rangle_t = \frac{1}{\tau_{\text{obs}}} \sum_{\tau=1}^{\tau_{\text{obs}}} A(\Gamma(\tau))$$

and because an MD simulation often fluctuates strongly in the beginning, we skip the first, say, 100 time steps:

$$A_{\rm obs} = \frac{1}{\tau_{\rm obs} - 100} \sum_{\tau = 101}^{\tau_{\rm obs}} A(\Gamma(\tau))$$

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**Relation between simulations and statistical physics** 

- In MD a time average gives the experimental quantity A.
  - However: in statistical physics we use ensembles
    - a set of points  $\Gamma\,$  in phase space
    - the likelihood of system being in the  $d\Gamma$  neigborhood of point  $\Gamma$  is given by the probability distribution  $\rho(\Gamma)d\Gamma$
  - $\rho(\Gamma)$  depends on external conditions: (constant) NVE, NVT, NPT:

e.g. with 
$$\rho_{NVE}(\Gamma)$$

or generally, for any ensemble,  $\rho_{ens}(\Gamma).$ 

- In statistical physics the time average is replaced by an ensemble average (why?)
  - go through all the points  $\{q_i, p_i\}$  in the ensemble phase space.
- In a Monte Carlo simulations the time average is replaced by going through a large set of points in phase space (using a Markov chain):

$$A_{\text{obs}} = \langle A \rangle_{\text{ens}} = \sum_{i=1}^{N_{\text{sim}}} A(\Gamma_i) \rho_{\text{ens}}(\Gamma_i)$$

• If  $\rho_{ens}(\Gamma)$  is independent of time (thermodynamic equilibrium), and the system is ergodic

$$\langle A \rangle_t = \langle A \rangle_{\text{ens}}$$

## Ergodicity

- In an ergodic system a long enough simulation will go through all points in phase space  $\{q_i, p_i\}$ .
- An example of a *non-ergodic* system (each hexagon represents one point phase space {q<sub>i</sub>, p<sub>i</sub>}):
  - In the darker area, the simulation moves in a close path, and can never get out of this area ⇒ the simulation does not test all of phase space, i.e. is non-ergodic.
  - In case there would be a single path which would go through the whole system, the system would be ergodic.
- Is it possible to prove that some system is ergodic? Not in the general case, and even for a given system it is usually very difficult in practice.



 In practice the system may not only have regions which are impossible to reach, but also regions which are surrounded by a high potential energy barr

which are surrounded by a high potential energy barrier so that reaching them in a finite simulation may be very unlikely (such a barrier is illustrated by the grey thin regions in the figure). This may distort the simulation averages badly.

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# Ergodicity

- A practical example:
  - Simulate diffusion in Cu at high temperature, around the melting point. In equilibrium the lattice has, say, 10 vacancies which cause diffusion at a rate of e.g. 1 atom/1 ps. Hence in a 100 ps simulation one gets about 1000 atom jumps, which appears to give a good time average of the diffusion constant.

**But:** about once in a ns a Frenkel pair, that is a pair of one vacancy and one atom at an interstitial position, may be created. Because the interstitial moves very much faster than the vacancy, it can cause thousands of atom jumps before it recombines with some vacancy. Because the interstitial causes a huge lot of diffusion, its presence can completely change the diffusion constant which would have been obtained in 100 ps.

So the system must be simulated for tens of ns's to get a reliable estimate of the diffusion coefficient - and if one does not realize the possibility of Frenkel pair formation, one would probably never notice this in a single 100 ps simulation. [Nordlund and Averback, *Phys. Rev. Lett.* **80** (1998) 4201]

• To get reliable results one not only has to burn away computer time, but also understand the physics in the system well!

## Ergodicity

• Sometimes (in MC simulations) it is useful to use a weighting function  $w_{ens}(\Gamma)$  to weight the ensemble and speed up getting the desired results:

$$\rho_{ens}(\Gamma) = \frac{w_{ens}(\Gamma)}{Q_{ens}}$$

$$Q_{ens} = \sum_{\Gamma} w_{ens}(\Gamma) \quad \text{(partition function)}$$

$$\langle A \rangle_{ens} = \frac{\sum_{\Gamma} w_{ens}(\Gamma)A(\Gamma)}{\sum_{\Gamma} w_{ens}(\Gamma)}$$

- MC integration: the flatter the function, the faster it is to obtain a precise average
- $Q_{\rm ens}$  will depend on the macroscopic properties of the system.
- Connection to thermodynamics:  $\Psi_{ens} = -\ln Q_{ens}$  = thermodynamic potential
- In practice: set up the MC simulation Markov chain such that it generates points according to the desired weighting function.
  - A simple choice:  $w_{ens}(\Gamma) = \rho_{ens}(\Gamma)$
  - · How this is achieved in practice will be dealt with in the MC course.

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## **Ergodicity**

- · So, to summarize the purpose of equilibrium simulations can be stated as:
  - go through phase space as efficiently as possible to get averages which correspond to experimentally
    observable quantities A<sub>obs</sub>
  - molecular dynamics:  $\langle A \rangle_t$
  - Monte Carlo:  $\langle A \rangle_{ens}$  (importance sampling)
- In MD only the NVE ensemble is obtained by solving the ordinary Newton/Lagrange/Hamiltonian equations of motion. For the other ones, one has to generate equations of motion which behave according to the desired ensemble  $\rho_{ens}(\Gamma)$

### The most important ensembles

[source: L.E. Reichl, A Modern Course in Statistical Physics]

- As in thermodynamics, the ensembles are denoted by letters which indicate which physical quantities are conserved. The names are also the same.
  - 1. Microcanonical (NVE)
  - 2. Canonical (NVT)
  - 3. Isothermal-isobaric (NPT)
  - 4. Grand canonical (µVT)
- Here *N* is the number of atoms, *V* the system volume, *T* the temperature, *P* the pressure, and μ the chemical potential [cf. e.g. Mandl "Statistical physics" chapters 2 and 11].
- Microcanonical: NVE constant (isolated)

$$\rho_{NVE}(\Gamma) = \delta(H(\Gamma) - E)$$
$$Q_{NVE} = \sum_{\Gamma} \delta(H(\Gamma) - E) = \frac{1}{N!} \frac{1}{h^{3N}} \int d\mathbf{r} d\mathbf{p} \delta(H(\mathbf{r}, \mathbf{p}) - E)$$

- Thermodynamical potential is the entropy:  $\frac{S}{k_{\rm B}} = \ln Q_{NVE}$ .
- The  $\delta$  function selects the states  $\Gamma$  where the total energy = *E*.
- Natural for MD in the sense that the total energy is conserved.

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## The most important ensembles

Canonical: NVT constant (closed but not heat-isolated)

 $\rho_{NVT}(\Gamma) \propto \exp(-H(\Gamma)/k_{\rm B}T)$ 

$$Q_{NVT} = \sum_{\Gamma} \exp(-H(\Gamma)/k_{\rm B}T) = \frac{1}{N!} \frac{1}{h^{3N}} \int d\mathbf{r} d\mathbf{p} \exp(-H(\mathbf{r},\mathbf{p})/k_{\rm B}T)$$

- Thermodynamical potential is the Helmholtz free energy:

$$\frac{A}{k_{\rm B}T} = -\ln Q_{NVT}, \quad A = E - ST$$

## Isothermal-isobaric: NPT constant

 $\rho_{NPT}(\Gamma) \propto \exp((-H(\Gamma) + PV)/k_{\rm B}T)$ 

$$Q_{NPT} = \sum_{\Gamma} \exp((-H(\Gamma) + PV)/k_{\rm B}T) = \frac{1}{N!} \frac{1}{h^{3N} V_0} \int d\mathbf{r} d\mathbf{p} \exp((-H(\mathbf{r},\mathbf{p}) + PV)/k_{\rm B}T)$$

Thermodynamical potential the Gibbs free energy:

$$\frac{G}{k_{\rm B}T} = -\ln Q_{NPT}, \quad G = E - TS + PV$$

• In MD the volume has also to be made variable.





### The most important ensembles

#### Grand canonical: μVT constant

$$\rho_{\mu VT}(\Gamma) \propto \exp((-H(\Gamma) + \mu N)/k_{\rm B}T)$$

$$\mathcal{Q}_{\mu VT} = \sum_{\Gamma, N} \exp((-H(\Gamma) + \mu N)/k_{\rm B}T) = \sum_{N} \frac{1}{N!} \frac{1}{h^{3N}} \exp(-\mu N/k_{\rm B}T) \int d\mathbf{r} d\mathbf{p} \exp(-H(\mathbf{r}, \mathbf{p})/k_{\rm B}T)$$

• Thermodynamic potential is the grand potential:

$$\frac{-\Omega}{k_{\rm B}T} = -\ln Q_{\mu VT}, \quad \Omega = E - TS - \mu N = -PV$$

- Now the number of atoms is changing: we have to have an algorithm to add or remove particles [not trivial in most practical (condensed matter) systems].
- In the thermodynamic limit (system size N→∞) all the ensembles are equivalent (but the fluctuations around the average may not be).

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## Calculating thermodynamical quantities

• Internal energy, that is, total energy (in the mdmorse code Etot):

$$E = \langle H \rangle = \langle K \rangle + \langle U \rangle = \langle \sum_{i} \frac{|\mathbf{p}_{i}|^{2}}{2m_{i}} \rangle + \langle U(\mathbf{q}) \rangle$$

- $U(\mathbf{q})$  is obtained directly from the potential energy calculation.
- Temperature

$$E_{kin} = \langle K \rangle = \frac{3}{2}Nk_{\rm B}T \qquad \Rightarrow \qquad T = \frac{2K}{3Nk_{\rm B}} = \frac{1}{3Nk_{\rm B}} \sum_{i=1}^{N} \frac{\left|\mathbf{p}_{i}\right|^{2}}{m_{i}}$$

• So, on the average there is  $k_{\rm B}T/2$  of energy per degree of freedom, as the classical equipartition theorem predicts.

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"particle reservoir"		

- Pressure (refer to Hamiltonian equations of motion):
  - · Generalized equipartition theorem for atom positions:

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$$\langle q_k \frac{\partial H}{\partial q_k} \rangle = k_{\rm B} T \implies -\frac{1}{3} \langle \sum_{i=1}^{N} \mathbf{r}_i \cdot (\nabla_{\mathbf{r}_i} U) \rangle = \frac{1}{3} \langle \sum_{i=1}^{N} \mathbf{r}_i \cdot \mathbf{f}_i^{\rm tot} \rangle = -N k_{\rm B} T;$$

• Divide the force into two components:  $\mathbf{f}_i^{\text{tot}} = \mathbf{f}_i^{\text{ext}} + \mathbf{f}_i$ Ν

external pressure: 
$$\frac{1}{3} \langle \sum_{i=1}^{N} \mathbf{r}_{i} \cdot \mathbf{f}_{i}^{\text{ext}} \rangle = -PV$$

intern

hal virial: 
$$W = -\frac{1}{3}\sum_{i=1}^{n} \mathbf{r}_{i} \cdot (\nabla_{\mathbf{r}_{i}}U) = \frac{1}{3}\sum_{i=1}^{n} \mathbf{r}_{i} \cdot \mathbf{f}_{i}$$

$$\Rightarrow \frac{1}{3} \langle \sum_{i=1}^{N} \mathbf{r}_{i} \cdot \mathbf{f}_{i} \rangle + \frac{1}{3} \langle \sum_{i=1}^{N} \mathbf{r}_{i} \cdot \mathbf{f}_{i}^{\text{ext}} \rangle = -Nk_{\text{B}}T \text{ which can be rewritten } \langle W \rangle - PV = -Nk_{\text{B}}T$$

N

 $\Rightarrow$  desired pressure  $PV = Nk_{B}T + \langle W \rangle$ 

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## Calculating thermodynamical quantities

• Pair interaction V(r) and periodic boundaries:

$$W = -\frac{1}{3} \sum_{i} \sum_{j>i} w(r_{ij}); \qquad w(r) = r_{ij} \frac{d}{dr_{ij}} V(r_{ij});$$

• Calculation in the force routine:

! dVdr is the derivative of V, i.e. the force virial=virial+dVdr\*(dx/r\*dx+dy/r\*dy+dz/r\*dz)

- · Thermodynamic potentials (free energies)
  - · Quantities which depend on the entropy
  - Energy/potential differences can be 'easily' calculated by integrating over a reversible path:

T

$$\left(\frac{A}{Nk_{\rm B}T}\right)_2 - \left(\frac{A}{Nk_{\rm B}T}\right)_1 = \int_{\beta_1}^{\beta_2} \left(\frac{E}{Nk_{\rm B}T}\right) \frac{d\beta}{\beta} = -\int_{T_1}^{T_2} \left(\frac{E}{Nk_{\rm B}T}\right) \frac{dT}{T} \\ \left(\frac{A}{Nk_{\rm B}T}\right)_2 - \left(\frac{A}{Nk_{\rm B}T}\right)_1 = \int_{\rho_1}^{\rho_2} \left(\frac{PV}{Nk_{\rm B}T}\right) \frac{d\rho}{\rho} = -\int_{V_1}^{V_2} \left(\frac{PV}{Nk_{\rm B}T}\right) \frac{dV}{V}$$

- · So one has to calculate a thermodynamic average for a large number of intermediate steps, then integrate over the path.
- · Calculating absolute values with the Frenkel-Ladd method: • Construct a potential energy which is dependent on a parameter  $\lambda$ :  $U = U(\mathbf{r}, \lambda)$

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## Calculating thermodynamical quantities

$$\Rightarrow \frac{\partial A}{\partial \lambda} = -k_{\rm B} T \frac{\partial}{\partial \lambda} \Big[ \ln \int d\mathbf{r} \exp(-U(\mathbf{r}, \lambda) / k_{\rm B} T) \Big]$$
$$= \frac{\int d\mathbf{r} \frac{\partial V}{\partial \lambda} \exp(-U / k_{\rm B} T)}{\int d\mathbf{r} \exp(-U / k_{\rm B} T)}$$
$$= \langle \frac{\partial U}{\partial \lambda} \rangle$$

- Construct U so that for  $\lambda = \lambda_0$  the absolute value of A can be calculated analytically or numerically: e.g. an ideal gas or a harmonic lattice.
  - Then get the absolute value of A for any  $\lambda$  using:

$$A(\lambda) - A(\lambda_0) = \int_{\lambda_0}^{\lambda} \langle \frac{\partial U}{\partial \lambda} \rangle d\lambda$$

- Real potential function, for which we want A , is  $U_0$ 
  - construct  $U = U(\mathbf{r}, \lambda)$  to interpolate between  $U_0$  and a harmonic lattice (Einstein's model) with

$$U(\mathbf{r}, \lambda) = U_0(\mathbf{r}) + \lambda \sum_{i=1}^{N} (\mathbf{r}_i - \mathbf{r}_{i0})^2$$
  
$$\Rightarrow \quad A(\lambda = 0) = A(\lambda) - \int_{0}^{\lambda} \langle \frac{\partial U}{\partial \lambda} \rangle d\lambda'$$

- At large values of  $\lambda$  we have harmonic lattice: e.g. Helmholtz free energy is:

$$A(\lambda) = \frac{3N\hbar\omega}{2} - 3Nk_{\rm B}T\ln(1 - e^{-\hbar\omega/k_{\rm B}T}) + O(1/\lambda)$$

and hence the free energy for our 'real' system  $U_0$  is  $A(\lambda = 0)$  and can be calculated by integrating over  $(\partial U)/(\partial \lambda)$ . [Frenkel-Ladd, *J. Chem. Phys.* **81** (1984) 3188]

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## Calculating thermodynamical quantities

#### Response functions

- · How does the system react to a change in some thermodynamic variable?
- Some of the most important response functions:

constant volume heat capacity
$$C_V = \left(\frac{\partial E}{\partial T}\right)_V$$
constant pressure heat capacity $C_P = \left(\frac{\partial H}{\partial T}\right)_P$ thermal expansion coefficient $\alpha_P = V^{-1} \left(\frac{\partial V}{\partial T}\right)_P$ isothermal compressibility $\beta_T = -V^{-1} \left(\frac{\partial V}{\partial P}\right)_T$ bulk modulus $B = 1/\beta_T$ thermal pressure coefficient $\gamma_V = \left(\frac{\partial P}{\partial T}\right)_V$ 

• Because  $\alpha_P = \beta_T \gamma_V$  it is enough to get one of these three coefficients

- How can one get these from simulations?
  - Direct simulation
    - E.g. heat capacity  $C_V$  can be obtained by doing simulations at different temperatures, thus obtaining  $E(T) \Rightarrow$

$$C_V(T) = \left(\frac{\partial E}{\partial T}\right)_V$$

- From the fluctuations in the system.
  - E. g.  $C_V$  from a single simulation in the canonical ensemble:

 $\langle \delta H^2 \rangle_{NVT} = k_B T^2 C_V$  (*H* is the momentaneous enthalpy)

• Because  $\langle \delta K \delta U \rangle_{NVT} = 0$ ,  $C_V$  can be separated into a kinetic and potential energy part:

$$\langle \delta H^2 \rangle_{NVT} = \langle \delta U^2 \rangle_{NVT} + \langle \delta K^2 \rangle_{NVT}$$

• Kinetic energy part:  $\langle \delta K^2 \rangle_{NVT} = \frac{3N}{2} (k_{\rm B}T)^2 = \frac{3N}{2\beta^2} \Rightarrow$  ideal-gas heat capacity  $C_V^{\rm id} = \frac{3}{2} N k_{\rm B}^2$ .

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### **Calculating thermodynamical quantities**

· By combining these we get

$$\langle \delta U^2 \rangle_{NVT} = k_{\rm B} T^2 \left( C_V - \frac{3}{2} N k_{\rm B} \right)$$

- So we can calculate  $C_V$  solely from the fluctuations of the potential energy.
- Similar fluctuation identities can also be derived for many other response functions (see e.g. Allen-Tildesley chapter 2.5.)
- These identities really depend on the ensemble used. E.g. in the microcanonical ensemble:

$$\langle \delta K^2 \rangle_{NVE} = \langle \delta U^2 \rangle_{NVE} = \frac{3}{2} N k_{\rm B}^2 T^2 \left( 1 - \frac{3Nk_{\rm B}}{2C_V} \right)$$

#### Structural quantities

Pair correlation function

$$g_2(\mathbf{r}_i, \mathbf{r}_i) = g_2(r_{ii}) = g(r)$$

which tells at what distances atoms are from each other.

· It can be calculated as

$$g(r) = \rho^{-2} \langle \sum_{i} \sum_{j \neq i} \delta(\mathbf{r}_{i}) \delta(\mathbf{r}_{j} - \mathbf{r}) \rangle$$
$$= \frac{N^{2}}{V^{2}} \langle \sum_{i} \sum_{i \neq j} \delta(\mathbf{r} - \mathbf{r}_{ij}) \rangle$$

• g(r) gives information on the structure of the material. For instance melting:



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Calculating thermodynamical quantities

• In practice it is of course not handy to use a delta function on a computer. So what is done instead is to collect statistics of what atom distances exist in some finite interval  $\Delta r$ :

```
integer :: stat(0:10000)
do i=0,10000
   stat(i) = 0
enddo
binwidth=0.01
do i=1.N
   do j=1,N
    if (i==j) cycle
    dx=x(j)-x(i)
    dy=y(j)-y(i)
    dz=z(j)-z(i)
    rsq=dx*dx+dy*dy+dz*dz
    r=sqrt(rsq)
    ir = int(r/binwidth+0.5)
    if (ir > 10000) ir=10000
    stat(ir) = stat(ir) + 1
   enddo
enddo
```

- Note: no boundary condition checks.
- The normalization factor  $4\pi r^2 \Delta r$  can be added afterwards, when printing the statistics.
- In practice if N is small (say 100 or less) the statistics will be poor  $\Rightarrow$  time averaging.

• g(r) is also useful because the average of any pair function can be given in the form:

$$\langle a(\mathbf{r}_i, \mathbf{r}_j) \rangle = \frac{1}{V} \int d\mathbf{r}_i d\mathbf{r}_i g(\mathbf{r}_i, \mathbf{r}_j) a(\mathbf{r}_i, \mathbf{r}_j) \text{ Or}$$
  
$$\langle A \rangle = \langle \sum_i \sum_{j>i} a(r_{ij}) \rangle = \frac{1}{2} N \rho \int_0^\infty a(r) g(r) 4 \pi r^2 dr$$

• E.g. the energy (pair interaction V(r)) $E = \frac{3}{2}Nk_{\rm B}T + 2\pi N\rho \int_{0}^{\infty} V(r)g(r)r^{2}dr$ or the pressure  $PV = Nk_{\rm B}T - \frac{2}{3}\pi N\rho \int_{0}^{\infty} w(r)g(r)r^{2}dr$ 

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Calculating thermodynamical quantities

• Structure factor in reciprocal k-space (Fourier transformation of positions):

$$\rho(k) = \sum_{i=1}^{N} \exp(i\mathbf{k} \cdot \mathbf{r})$$

• The square of  $\rho(k)$  gives the structure factor S(k):

$$S(k) = N^{-1} \langle \rho(k) \rho(-k) \rangle,$$

which can be measured with x-ray or neutron scattering

• This quantity can be shown to be related to g(r) through a 3-dimensional Fourier transform:

$$S(k) = 1 + \rho \hat{g}(k) = 1 + 4\pi \rho \int_{0}^{\infty} \frac{\sin kr}{kr} g(r) r^{2} dr$$

- Because g(r) is a measurable quantity, it is often useful in testing how realistic a potential energy function is in describing some structure, especially a liquid or amorphous phase.
- However, this test is actually not all that sensitive to the detailed structure.

- Transport coefficients
  - The correlation between any two quantities A and B is

$$\begin{split} c_{AB} &= \frac{\langle \delta A \delta B \rangle}{\sigma(A)\sigma(B)} \\ \sigma^{2}(A) &= \langle \delta A^{2} \rangle = \langle A^{2} \rangle - \langle A \rangle^{2} \\ \delta A &= A - \langle A \rangle \\ 0 &\leq c_{AB} \leq 1 \end{split}$$

- The time dependent correlation function  $c_{AB}(t)$ : A and B at different times, e.g. A(t) and B(0)
- Autocorrelation function  $c_{AA}(t)$

• Correlation time 
$$t_A = \int_{0}^{\infty} c_{AA}(t) dt$$

- These give information on
- the dynamics of the material
- transport coefficients
- can be related to experimental spectra by Fourier transformations

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## Calculating thermodynamical quantities

- Transport coefficients: system response to an external disturbance  $\rho(t) = \rho_{ens} + \delta \rho(t)$ 
  - For instance diffusion coefficient: particle flux  $\leftrightarrow$  concentration gradient.
  - $\rho(t) \rightarrow$  time dependent averages.
  - Comparison to transport equations  $\rightarrow$  transport coefficients.

• Coefficients usually of the form 
$$\gamma = \int_{0}^{\infty} \langle \dot{A}(t)(\dot{A}(0)) \rangle dt$$

· For a large time there also always exists an Einstein relation

$$2t\gamma = \langle (A(t) - A(0))^2 \rangle$$

· Some transport coefficients for the NVE-ensemble:

Diffusion constant	$D = \frac{1}{3} \int \langle \mathbf{v}_i(t) \cdot \mathbf{v}_i(0) \rangle dt$
Simple form to evaluate:	$2tD = \frac{1}{3} \langle (\mathbf{r}_i(t) - \mathbf{r}_i(0))^2 \rangle$
Thermal conductivity	$\lambda_T = \frac{V}{k_{\rm B} T^2} \int_0^\infty \langle j_i^{\varepsilon}(t) j_i^{\varepsilon}(0) \rangle dt,$
	$2t\lambda_T = \frac{V}{k_{\rm B}T^2} \langle (\delta \varepsilon_{\alpha}(t) - \delta \varepsilon_{\alpha}(0))^2 \rangle$ , where
	$\delta \varepsilon_{\alpha} = \frac{1}{V} \sum_{i} r_{i\alpha} (\varepsilon_{i} - \langle \varepsilon_{i} \rangle);$
	$j_i^{\varepsilon} = \frac{\partial \delta \varepsilon}{\partial t}^{\alpha};$
	$\varepsilon_i = \frac{p_i^2}{2m_i} + \frac{1}{2} \sum_{i \neq j} V(r_{ij})$

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### Algorithms for simulating ensembles

[most material from Allen-Tildesley ch. 7.4]

- Pure NVE: see lectures 2-5
- NVE-scaling or constraint methods:
  - Often even in an NVE simulation one does some simple tricks to control temperature and/or pressure. This gives something of an NVT or NVP and NVE hybrid: *T* and *P* fluctuate, and the system does not behave as a true NVT or NVP ensemble in the thermodynamic sense. But on average *T* and *P* have the desired value. In true NVT or NPT algorithms it is possible to have *T* and *P* have exactly the desired value, and the simulation directly corresponds to the thermodynamic ensembles.

#### Temperature scaling

• Trivial scaling: force during every time step the system temperature to be exactly *T*. This may be a rather severe perturbation of the atom motion especially if there are only a few atoms. It suppresses normal *T* fluctuations, and does still not correspond to a true NVT ensemble. But the error in ensemble averages usually is O(1/N) so with a large number of atoms one may get away with it.

- The Berendsen method: essentially a direct scaling but softened with a time constant. [Berendsen et al. J. Chem. Phys. 81 (1984) 3684].
  - Coupling to heat bath, Langevin dynamics:  $m\dot{v} = F m\gamma v + R(t)$
  - Global coupling + local noise
  - Replace the local noise by its average behvior in  $\frac{\mathrm{d}E_{\mathrm{k}}}{\mathrm{d}t}$

$$\Rightarrow \left(\frac{\mathrm{d}T}{\mathrm{d}t}\right)_{\mathrm{bath}} = 2\gamma(T_0 - T)$$
$$\Rightarrow m\dot{v} = F + m\gamma\left(\frac{T_0}{T} - 1\right)v$$

• Let  $T_0$  be the desired temperature,  $\Delta t$  the time step of the system and  $\tau_T = 1/2\gamma$  the time constant of the control. In the Berendsen method in order to change the temperature in one timestep by  $2\gamma\Delta t(T_0 - T) = \frac{\Delta t}{\tau_T}(T_0 - T)$  all atom velocities are scaled at every time step with a factor  $\lambda$ , where

$$\lambda = \sqrt{1 + \frac{\Delta t}{\tau_T} \left(\frac{T_0}{T} - 1\right)}$$
 (\*)

• Note: if  $\tau_T > 100 \Delta t$  then the system has natural thermal fluctuations about the average.

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### Algorithms for simulating ensembles

- The derivation above lacks a factor 2<sup>1</sup>.
- Let's write the temperature behavior as  $\frac{dT}{dt} = \frac{1}{\tau_T}(T_0 T)$ . From this we can solve T(t) as

$$T(t) = T_0 + (T_i - T_0)e^{-t/\tau_T}$$
, where  $T_i = T(0)$  is the initial temperature.

• On the other hand when we scale velocity  $v \rightarrow \lambda v$  the change in the internal energy is  $\delta E = (\lambda^2 - 1)\frac{3}{2}Nk_BT$ ,

• Now the heat capacity is  $C_V = \frac{\delta E}{\delta T}$ . From this and from the differential equation of the temperature we get

$$\frac{\delta T}{\delta t} = \frac{1}{\tau_T} (T_0 - T) \Rightarrow \delta T = \frac{\delta t}{\tau_T} (T_0 - T)$$

• From the definition of heat capacity we obtain  $C_V = \frac{\delta E}{\delta T} = \frac{(\lambda^2 - 1)\frac{3}{2}Nk_{\rm B}T}{\frac{\delta t}{\tau_T}(T_0 - T)}.$ 

• By solving  $\lambda$  from this we get  $\lambda^2 = \frac{2C_V \delta t}{3k_B N \tau_T} \left(\frac{T_0}{T} - 1\right) + 1$ .

• Let's make the bold assumption that the heat capacity is given by the Dulong-Petit law:  $C_V = 3Nk_{\rm R}$ 

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<sup>1.</sup> Ideas for this derivation are from Kalevi Kokko's lecture notes at http://vanha.physics.utu.fi/opiskelu/kurssit/XFYS4416/

• Finally we obtain the expression for  $\lambda$ :

$$\lambda^2 = \frac{2\delta t}{\tau_T} \left( \frac{T_0}{T} - 1 \right) + 1.$$
 (\*\*)

- As we shall see in exercise 7, this is the right expression in the sense that it reproduces the behavior dictated by the equation  $\frac{dT}{dt} = \frac{1}{\tau_T}(T_0 T)$ .
- Effect of parameter  $\tau_{T}$  on time development of T



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## Algorithms for simulating ensembles

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• ... and on T fluctuations



 $\star$ : simulation for  $\tau_T = \infty$ , i.e. no temperature control

#### • Pressure scaling (Berendsen)

Pressure is put to a desired value by changing the cell size.

- If the desired pressure is  $P_0$  and  $\tau_P$  is the time constant, the scaling factor is

$$\mu = \sqrt[3]{1 - \frac{\beta \Delta t}{\tau_P} (P_0 - P)}$$

where  $\beta$  is the isothermal compressibility of the system = 1/bulk modulus.

- $\beta$  only occurs in the division over the time constant  $\tau_P$  it is just a factor which makes the typical time constant values roughly independent of the material.
- Scaling implemented by changing all atom positions x and the system size S every time step

$$x(t + \Delta t) = \mu x(t)$$
  
$$S(t + \Delta t) = \mu S(t)$$

• Also the system volume V changes:

$$V(t + \Delta t) = \mu^3 V(t)$$

· Pressure scaling done after the solution of the equations of motion

• 
$$\tau_P > 100 \Delta t$$

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### Algorithms for simulating ensembles

• Another (better) way to derive  $\mu$ :

• We want 
$$\frac{\mathrm{d}P}{\mathrm{d}t} = \frac{1}{\tau_P} [P_0 - P(t)]$$
 (\*)

• Volume scaling  $V \rightarrow \mu^3 V$ . Definition of compressibility:  $\beta = -\frac{1}{V \partial P} \Rightarrow \frac{\partial P}{\partial V} = -\frac{1}{V \beta}$ .

• Now 
$$\frac{\mathrm{d}P}{\mathrm{d}t} = \frac{\mathrm{d}P}{\mathrm{d}V}\frac{\mathrm{d}V}{\mathrm{d}t} = -\frac{1}{V\beta}\frac{(\mu^3 - 1)V}{\Delta t} = \frac{1 - \mu^3}{\beta\Delta t}.$$

• From this and (\*) we get  $\frac{1-\mu^3}{\beta\Delta t} = \frac{1}{\tau_P} [P_0 - P(t)]$ , from which we solve  $\mu$ :

$$\mu^3 = 1 - \frac{\beta \Delta t}{\tau_P} [P_0 - P(t)]$$

- Effect of parameter  $\tau_P$
- The Berendsen scaling can be used to control *T* and *P*. If the system is in equilibrium the total energy *E* should still be conserved, but if phase transitions, such as melting occur, *E* does not necessarily stay conserved until equilibrium is reached again.
- In the Berendsen method P, T, V and  $E_{\text{pot}}$  all fluctuate, and because the time constants  $\tau$  have to be fairly large

it can take quite a while to reach a desired pressure or temperature.



• But in equilibrium and with large enough time constants, the method gives quite realistic fluctuations in *T* and *P*. And it is almost as trivial to implement as direct scaling. Hence it is much to be preferred over direct scaling.

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Algorithms for simulating ensembles

- True NVT algorithms
- The Andersén method [H. C. Andersén, J. Chem. Phys. 72, 2384 (1980)].
  - Give the atom with some probability a new velocity which corresponds to a desired heat bath temperature  $T_0$
  - · Physical interpretation clear: connection to external heat bath
  - Suitable for calculating thermodynamic averages, but not for looking at atomic processes in detail, since the random velocity is obviously an unphysical perturbation on the motion of a single atom.
- Nosé-Hoover-method [W. Hoover, Phys. Rev. A 31, 1695-1697 (1985).]
  - A fictional degree of freedom *s* which has its own kinetic and potential energy is added to the system, and this degree of freedom controls the temperature. The system total energy, i.e. Hamiltonian:

$$H = \sum_{i} \frac{\boldsymbol{p}_i}{2m_i} + V(\boldsymbol{q}_i) + \frac{Q}{2}p_s^2 + qkT \ln s$$

where  $p_s$  is the momentum associated with the degree of freedom.

• Now the Hamiltonian equations of motion become:

$$\frac{d\boldsymbol{q}_i}{dt} = \frac{\boldsymbol{p}_i}{m_i}; \qquad \frac{d\boldsymbol{p}_i}{dt} = -\frac{dV}{d\boldsymbol{q}_i} - p_s \boldsymbol{p}_i, \qquad \frac{dp_s}{dt} = \left(\sum_i \frac{\boldsymbol{p}_i}{m_i} - gkT\right)/Q$$

- These can be solved with some suitable algorithm.
- *Q* is a fictional mass related to the extra degree of freedom, which describes the rate at which the temperature changed.
  - Nosé suggested  $Q \sim gk_B T$  where g is the number of degrees of freedom in the system, typically 6N. For large Q the connection to the heat bath weakens, and for small Q the energy E may oscillate too much.
- Nosé-Hoover chains [Tobias, Martyna, Klein, J. Phys. Chem. 97 (1993) 12959]
  - Also control the new degree *s* of freedom with another Nosé-Hoover-algorithm and so forth, i.e. form a chain of these.
  - At least in simulations of proteins this can give a very good temperature control.
  - "Massive" Nosé-Hoover-chain: add a Nosé-Hoover thermostat chain to every degree of freedom (!)
    Advantage: as Nosé-Hoover, but in addition very efficient in equipartitioning the energy and thus getting the system into equilibrium. Disadvantage: even more coding

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### Algorithms for simulating ensembles

- True NPT algorithms
  - Andersén pressure control [H. C. Andersén, J. Chem. Phys. 72, 2384 (1980)]
    - The cell size V a dynamic variable, but the system shape may not change. The size is controlled by a fictional piston which has a mass Q (in units of m/l<sup>4</sup>). The kinetic and potential energy of the piston are:

$$E_{\operatorname{kin} V} = \frac{1}{2}Q\dot{V}^2$$
 and  $E_{\operatorname{pot} V} = PV$ 

and if the atom positions **r** and velocities **v** are written in reduced units *s* such that  $r = V^{1/3}s$  and  $v = V^{1/3}\dot{s}$  we get the equations of motion

$$\ddot{s} = \frac{f}{mV^{1/3}} - \frac{2}{3}\dot{s}\frac{\dot{V}}{V}$$

$$\ddot{V} = \frac{P_t - P}{Q}$$

where f are the forces acting on atoms,  $P_t$  is the momentaneous pressure and P the desired pressure.

- Parrinello-Rahman-pressure control [Parrinello and Rahman, J. Appl. Phys. 52 (1981) 7182]
  - This method also allows a variable simulation cell shape, that is, the angles between the axes do not have to be 90°.
  - The cell size and shape is given by vectors **a**, **b** and **c**. If we form a  $3 \times 3$ -matrix **h** out of these the atom positions **r** can be written in the form

 $\mathbf{r} = \mathbf{hs}$ .

where  $\ensuremath{s}$  is an ordinary vector.

• The equations of motion can be derived to be:

$$\ddot{\mathbf{s}}_{i} = -\sum_{j} \frac{dV}{dr} \frac{1}{m_{i}r_{ij}} (\mathbf{s}_{i} - \mathbf{s}_{j}) - \frac{\dot{\mathbf{G}}}{\mathbf{G}} \dot{\mathbf{s}}$$
$$W\ddot{\mathbf{h}} = (\mathbf{P} - p\mathbf{I})\boldsymbol{\sigma} - \mathbf{h}\boldsymbol{\Sigma}$$

where  $\mathbf{G} = \mathbf{h}^T \mathbf{h}$ ,  $\sigma$  is a tensor which defines reciprocal space, and  $\mathbf{P}$  is the generalized  $3 \times 3$  pressure tensor:

$$\mathbf{P} = \frac{1}{V} \left[ \sum_{i} m_{i} \mathbf{v}_{i} \mathbf{v}_{i} - \sum_{i} \sum_{j > i} \frac{1}{r_{ij}} \frac{du}{dr_{ij}} \mathbf{r}_{ij} \mathbf{r}_{ij} \right]$$

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### Algorithms for simulating ensembles

- The diagonal elements of **P** are the pressures in x, y and z, the other elements are shear elements.
- The hydrostatic "ordinary" pressure  $P = (tr \mathbf{P})/3 = (P_{11} + P_{22} + P_{33})/3$ .
- $\Sigma$  is a quantity which depends on the external pressure tensor **S**:

$$\Sigma = \mathbf{h_0^{-1}}(\mathbf{S} - p)\mathbf{h_0^{T-1}}\Omega_{\mathbf{0}}$$

where  $\mathbf{h}_0~$  and  $\boldsymbol{\Omega}_0$  are the original (reference) shape and volume of the system.

• *W* is a fictional "mass" which is used to control the rate of change of the pressure (compare with *Q* in the NVT algorithms above).

• This allows us to simulate a system which changes shape, for instance a cubic to hexagonal phase transformation.



• An example of the effects of the mass parameter *Q*:A Lennard-Jones-system (Ne); T=0.1 K; constant pressure-MD:



himmän jakson pituus.

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## Algorithms for simulating ensembles

### • µVT-methods

- Chemical potential  $\mu$  stays constant, number of atoms fluctuates
- Rarely used in MD, more often in MC simulations where it is more natural to add and remove atoms from the system.
  - An alternative to adding or removing atoms is to add or remove "control volume".
- In condensed matter simulations the problem is that just adding an atom on a random place can easily lead to completely unphysical configurations.
- Also adding or removing control volume without distrorting the system state too much may be tricky.
- If you need this, see e.g. [Lynch, Pettitt: *J. Chem. Phys.* **107** (1997) 8594] or [Heffelfinger, *J. Chem. Phys.* **100** (1994) 7548].

- What T and P control to use?
- For *T* or *P* scaling: Berendsen is fast to implement, and does work well provided the time constants are large enough.
- If one wants accurate *T* control or needs to do NVT thermodynamic averaging, one of the Nosé-Hoover methods is probably best
- For orthogonal box NPT simulations: Andersén
- If one wants needs to deal with shear pressure or changes in crystal structure  $\Rightarrow$  Parrinello-Rahman

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### Other types of MD simulations

#### • Non-equilibrium MD (NEMD)

- Any MD simulation of a system which is not in thermodynamic equilibrium.
- Usually some perturbative term is added to the equations of motion.
- For instance for simulating viscosity, heat conductivity and atomic diffusion there are special NEMD algorithms.
- At its simplest, the perturbation can be an external force acting on some of the atoms.
- The external force heats the system up, which can be compensated by temperature control.

#### Brownian dynamics or Langevin dynamics

- Random forces are let to act on some atoms some of the time. This can be useful e.g. in speeding up infrequent events.
- This can also correspond to e.g. a large protein molecule in a liquid solvent. If the protein atoms do not react with the solvent atoms, and the solvent atoms are not interesting in themselves, their effect on the protein can be thought to reduce to random Langevin forces.

## Other types of MD simulations

#### Multiple time step methods

- In these methods the simulation is sped up by using different time steps for different atoms or parts of the system.
- A simple example of where this may be useful: a molecule which has light and much heavier particles. The light particles move much faster, so their motion can be simulated with a short time step  $\Delta t_1$  and the

heavy ones with a longer time step  $\Delta t_2$ .

• Another possibility: count near interactions acting on atom *i* with a short time step  $\Delta t_1$  and those farther away with a longer one  $\Delta t_2$ . In here, we assume the movement of the atoms far away is so small that they do not move significantly with respect to atom *i* during the shorter time  $\Delta t_1$ .

#### MD far from equilibrium

- Many processes of modern interest involve physical interactions which occur very far from thermodynamic equilibrium.
- E.g. two nanoparticles colliding in vacuum, or an energetic ion from an accelerator hits a material.
- In both cases very violent interactions occur over ps timescales, and the surrounding medium does not have time to
  equilibrate the system into anything close to thermodynamic equilibrium during the time when the interesting processes occur.
- Simulating such a system is simple: simply use ordinary NVE with no T or P scaling.
   But watch out for possible finite size effects!

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## Other types of MD simulations

- An example of NEMD: heat conduction in crystalline and amorphous Si [von Alfthan *et al.*, *MRS Symposium Proceedings*, 703 (2002) V6.2.1]
- Straightforward way: impose a T gradient  $\Rightarrow$  heat flux  $J \Rightarrow k = -J/\frac{dT}{dx}$ .
  - Problems: large fluctuations in  $J \Rightarrow$  large dT/dx needed.



## Other types of MD simulations

- Another way [Müller-Plathe, *J. Chem. Phys.* **106** (1997) 6082.]: impose heat flux by exchanging particle velocities between hot and cold parts of the system
- Flux 'exact', controlled by exchange interval
- dT/dx's for different simulation system sizes: (a)
  L<sub>c</sub> = 296 Å, L<sub>a</sub> = 100 Å,
  d = 32 Å
  (b) L<sub>c</sub> = 187 Å, L<sub>a</sub> = 38 Å,
  d = 16 Å
  (c) L<sub>c</sub> = 187 Å, L<sub>a</sub> = 38 Å,
  d = 32 Å

System	$k_{\mathrm{a}}$ (W/mk)	$k_{\rm c}^{}$ (W/mk)
(a)	0.93	13
(b)	0.85	9
(c)	0.80	15



- Results for c-Si size dependent, moreover experimental  $k_{\rm c} = 160 \text{ W/mK}$  !
- Phonon mean free path in c-Si ~ 1000 Å
- Results for  $k_a$  reasonable.
- No thermal boundary resistance observed.

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